

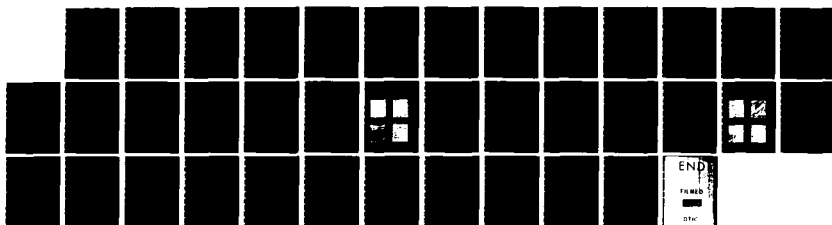
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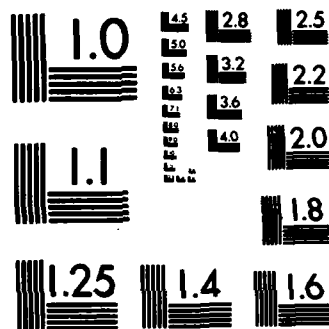
DURABILITY OF ADHESIVE BONDS BETWEEN GLASS OR METAL
SUBSTRATES AND A POLYBUTADIENE-POLYURETHANE(U) AKRON
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20. ABSTRACT (Continue on reverse side if necessary and identify by block number) The durability of polybutadiene-polyurethane joints to glass or metals in water at room and elevated temperatures was studied. The effects of silane pretreatment of the surface and of surface morphology on adhesion were examined. The durability of the joint was related to the number of chemical bonds that could form between the polyurethane and the surface and to the type of surface treatment given to the surface. Proper treatment led to significant enhancement of durability of the joints. The polyurethane was prepared from ARCO's hydroxyterminated polybutadiene (R-45HT), toluene		

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Technical Report No. 31

DURABILITY OF ADHESIVE BONDS BETWEEN GLASS OR
METAL SUBSTRATES AND A POLYBUTADIENE-POLYURETHANE

BY

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Adhesives are being used increasingly in engineering applications. One problem frequently encountered is that the mechanical properties of the bond component rapidly deteriorates upon exposure of the joint to its normal operating environment. Water (moisture) is one of the most hostile and common environments to which adhesive joints are exposed. This paper is concerned with the durability of polyurethane joints subjected to wet climate conditions. In order to accelerate such exposure the joints were immersed in water at elevated temperature. The effect of silanes and surface morphology of etched substrates on joint durability was examined.

Experimental

Materials

The materials used were 3-aminopropyltriethoxysilane (AS), (γ -glycidoxypopyl)trimethoxysilane (GS), 3-chloropropyltrimethoxysilane (CS), and 3-mercaptopropyltrimethoxysilane (MS) from Petrarch Systems Inc.; TDI (80/20 mixture of 2,4- and 2,6-isomers of toluene diisocyanates) from BASF Wyandotte Corporation; Isonal-100 (N,N-bis-(2-hydroxylpropylaniline) from Upjohn Polymer Chemicals; MICRO laboratory cleaner from International Product Corp.; and hydroxy-terminated polybutadene resin (R-45HT) from ARCO Chemical Co. The properties of R-45HT were listed in earlier studies.¹

Precleaned opticlear soda lime microscope slides (25 x 25 x 1.06 mm) were from KIMBLE. Mirror finish chrome ferro-type stainless steel places were from Appolo. The aluminum alloy was 300 3-H-14.

Preparation of Polyurethane

The polyurethane was prepared from hydroxyterminated polybutadiene (R-45HT), TDI and N,N-bis(2-hydroxypropyl) aniline. The ratio $[-NCO]/\text{total } [-OH]$ was kept constant and equal to 1. 6% excess $[-NCO]$ to hydroxyl number of R-45T was used. The equations for calculating the required amounts of diisocyanate and N,N-bis(2-hydroxypropyl) aniline and the procedure were the same as in previous studies¹.

Surface treatment of substrates

All amounts are in parts by weight unless otherwise stated.

Glass. Unabraded glass slides were heated at 140°C for 1 hr. and kept in a dessicator over $CaCl_2$ until treated with the desired reagent.

Surface abraded slides were ground with #600 aluminum oxide abrasive from BARNES Engineering Company, rinsed with distilled water, and immersed for 24 hrs. at room temperature in a mixture of MICRO laboratory cleaner with H_2O (50/50 by weight). The solution was heated to 90°C for 1 hr and then the slides were rinsed with distilled water, immersed in 90°C distilled water for 1 hr., rinsed with cold distilled water, dried at 140°C for 1 hr. and kept in a dessicator over $CaCl_2$.

Metals. Small plates (Approximately 75 x 25 x 1 mm) cut from the ferro-type plates were treated by one of four different methods before applying the chemical and elastomer layers:

1. Cleaned with petroleum ether or acetone and dried at 60°C for 40 minutes.
2. Cleaned as in 1; immersed for 10 minutes at 90°C in a solution

of concentrated hydrochloric acid (50 parts) and distilled water (50 parts); rinsed with cold distilled water; and dried at 60°C for 40 minutes.

3. Cleaned as in 1; immersed for 20 minutes at 80°C in a solution of oxalic acid (9 parts), sulfuric acid (1 part), and distilled water (80 parts); and rinsed and dried as in 2.
4. Cleaned as in 1; immersed at 70-75°C for 5 minutes in a solution of hydrochloric acid (83.3% by volume), phosphoric acid (12.5% by volume); and rinsed and dried as in 2.

The surfaces of aluminum plates of similar size were prepared by one of four methods:

5. Cleaned with petroleum ether and dried at 60°C for 40 minutes.
6. Abraded with fine sandpaper; and cleaned and dried as in 5.
7. Cleaned as in 5; immersed for 10 minutes in a 70°C solution of distilled water (30 parts), sulfuric acid (10 parts), sodium dichromate (1 part); and cleaned and dried as in 2.
8. Cleaned as in 5; immersed in 15% NaOH at 40°C for 5 minutes; rinsed with distilled water; and then method 7 without the petroleum ether step.

Application of silane solution

The silanes were applied at room temperature by immersing the substrates for 5 min. in a 0.5% by weight solution of the silane in a 50/50 mixture by volume of ethanol and water. Several drops of hydrochloric acid (pH_~4) as a catalyst for silane were added also except when the silane was AS. After removing the substrates from the silane solution, they were dried in air and then heated for 30 minutes at 100°C before applying the elastomer layer.

Application of elastomer layer and curing

For adhesion studies appropriately treated substrates were placed in a Teflon coated compression mold and the prepolymer containing the glycol was poured on. The top plate of the mold was covered with a sheet of washed, dried and pressed cotton cloth. The sample was cured at 100°C for 4 hrs. at 25 psi per 5 inch ram in a preheated PHI press. Rubber sheets were cured in a vertical mold in a Vacuum Oven at 100°C for 5 hrs.

Measurement of work of adhesion

180° peel tests were carried out on strips of cloth backed elastomer layer after trimming them to a uniform width of 2 cm. on the substrate. The cloth-backed elastomer layer was peeled off the substrate at a constant rate of 0.5 cm/min. The work of adhesion per unit area of interface W_A was calculated from the time average of the peel force P per unit width w of the detaching layer $W_A = 2P/w$.

Tensile tests

Tensile tests were carried out at room temperature and a cross head speed of 50 cm/min. using dumbbell specimens. Breaking elongation was calculated from the ratio $(L_b - L_o)/L_o$, where L_o and L_b were the initial length and the length at break, respectively.

Contact angle measurements

Contact angle measurements were carried out at room temperature with a contact angle goniometer manufactured by KERNCO Instrument Co. For each liquid five drops were placed on a substrate, and contact angle readings were taken from both left and

right sides of the liquid-air-solid interface. Readings were recorded at 10 and 20 minutes after the drops were placed on the substrate.

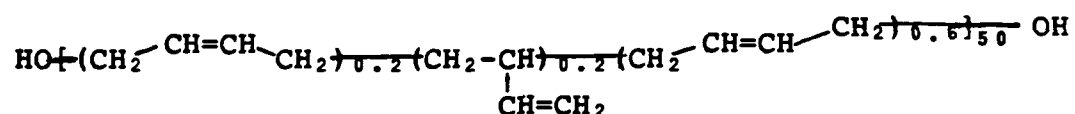
Electron microscopy

The electron microscopy was carried out with scanning electron microscope model JSM-U3.

Results and Discussion

Effect of water on bulk properties of polyurethane

According to ARCO hydroxyl terminated polybutadiene resin, R-45HT has the following chemical structure^{2,3}.



In contrast to the two principal classes of polyurethanes, from polyethers and from polyesters, the polyurethanes formed from polybd resin display certain properties and performance characteristics attributable to the hydrocarbon backbone. The most outstanding of these characteristics are outstanding low temperature properties and good hydrolytic stability. The glass transition temperature of the polyurethane derived from a prepolymer containing 6% Free NCO is about -65°C^1 . As shown in Fig. 1, when this polyurethane was immersed in distilled water at 70°C , the ultimate tensile strength σ_b , and the elongation at break, e_b , decreased slightly with increasing immersion time. After soaking for 55 days at 70°C σ_b decreased by about 30%, while e_b decreased by about 25%.

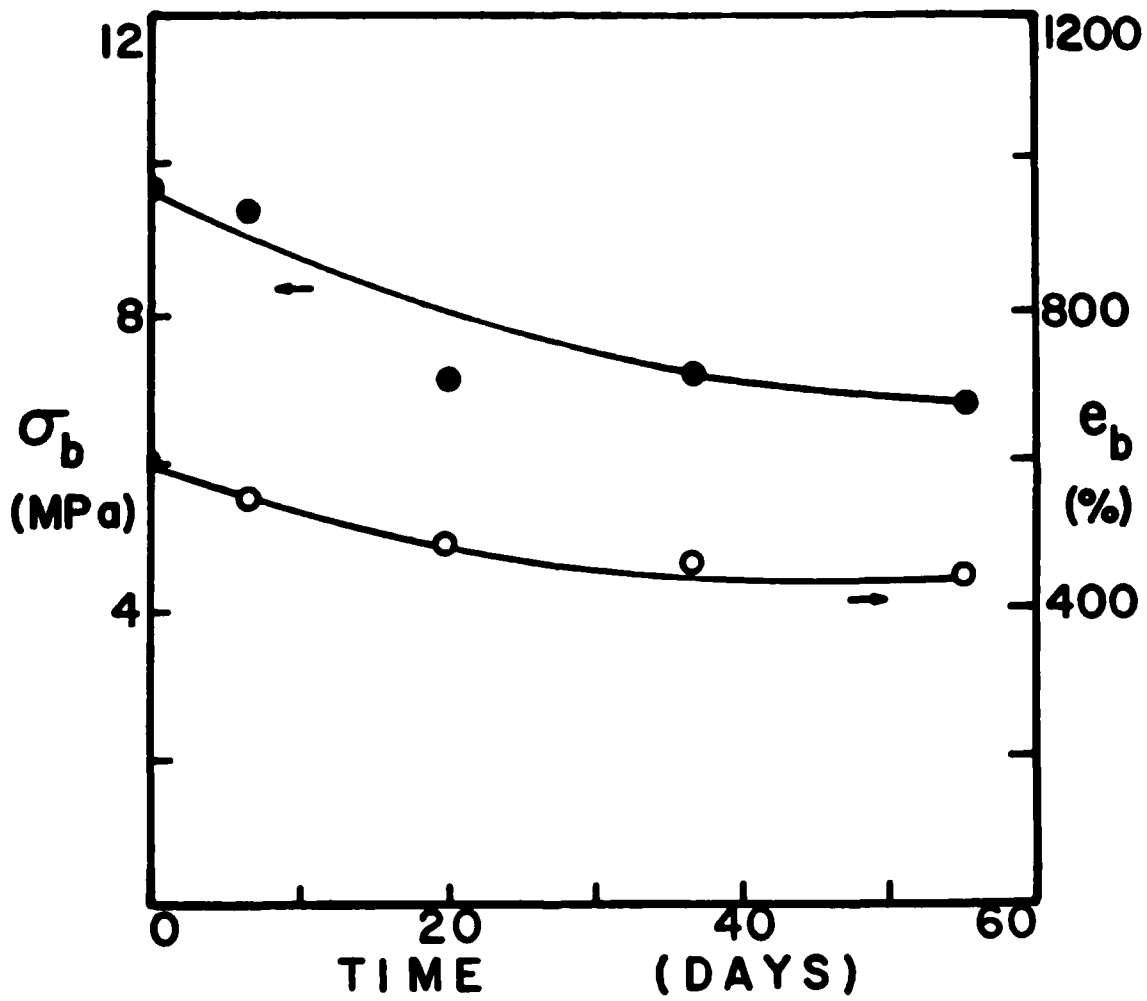
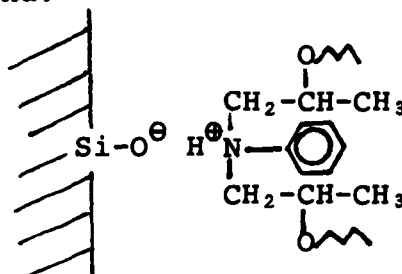


Fig. 1 - Effect of water on ultimate tensile strength (σ_b) and breaking elongation (e_b).

Adhesion of polyurethane to smooth glass surfaces

Warburg and Ihmori⁴ recognized that glass surfaces, which have been exposed to room air, are covered by a water film. Infra-red adsorption spectra of physically adsorbed water in porous silica show that the water is hydrogen bonded both intermolecularly and to hydroxyl groups on the silica surface⁵. When the polyurethane elastomer of this study is applied to a glass surface, the tertiary amine groups from the N,N-bis(2-hydroxypropyl)aniline in the backbone of the polyurethane can react with the acidic silanol groups present in the glass surface or perhaps even with surface moisture to form an ionic bond.



Our earlier studies indicated that high adhesion results. But as shown in Fig. 2 the work of adhesion fell dramatically, when glass/polyurethane peel test specimens were immersed in distilled water at room temperature. In order to improve the wet adhesion, glass slides were treated with AS, GS, or MS before applying the polyurethane. The dry adhesion increased remarkably. At equivalent concentrations of silanes the order of enhancement in the work of adhesion was AS > MS > GS. Cohesive failure was observed for glass slides pretreated with AS and MS. As in our earlier studies, results can be explained in terms of the different structures of silanes^{1, 6-9} AS has two reactive hydrogen atoms that can react

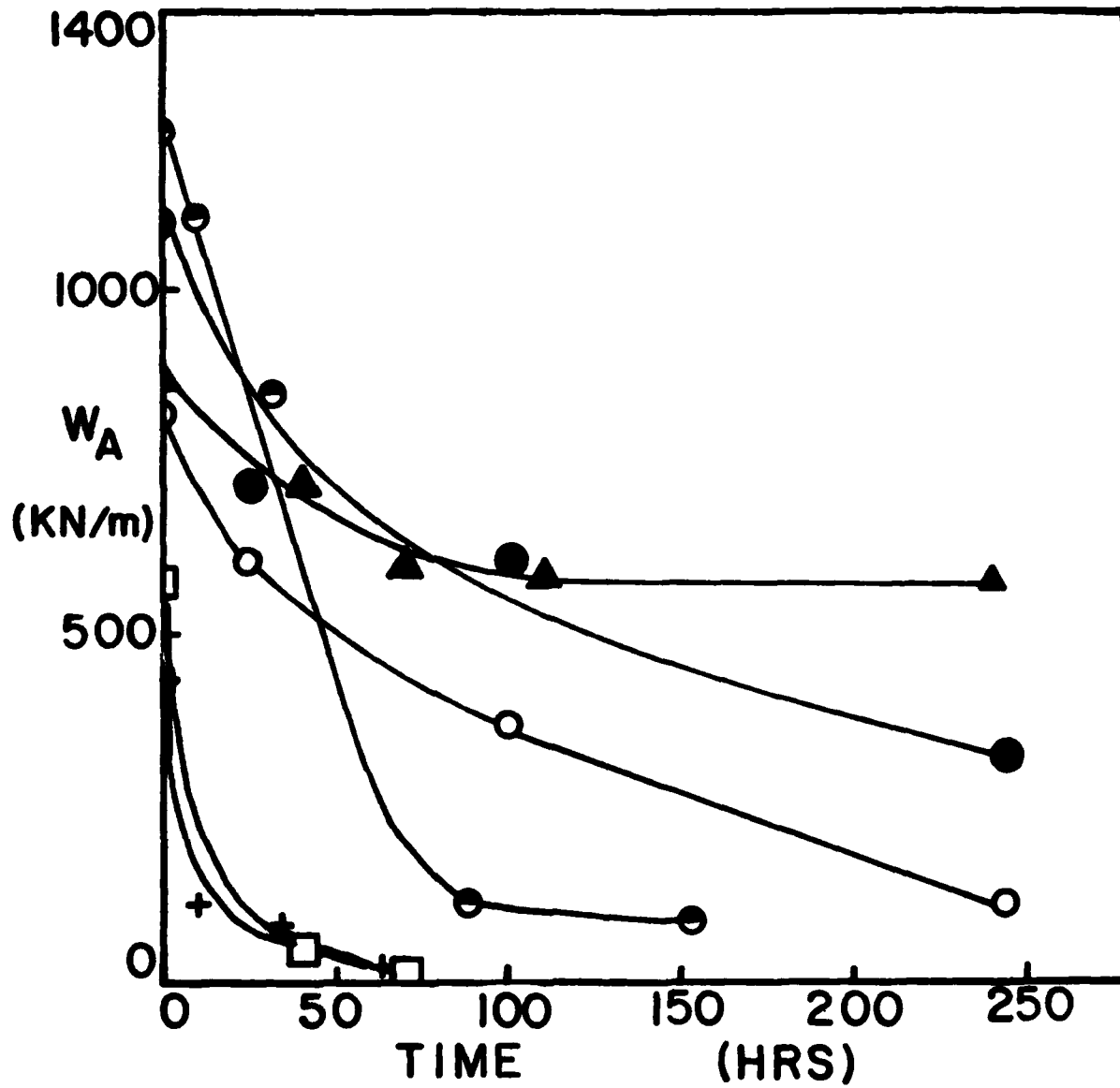
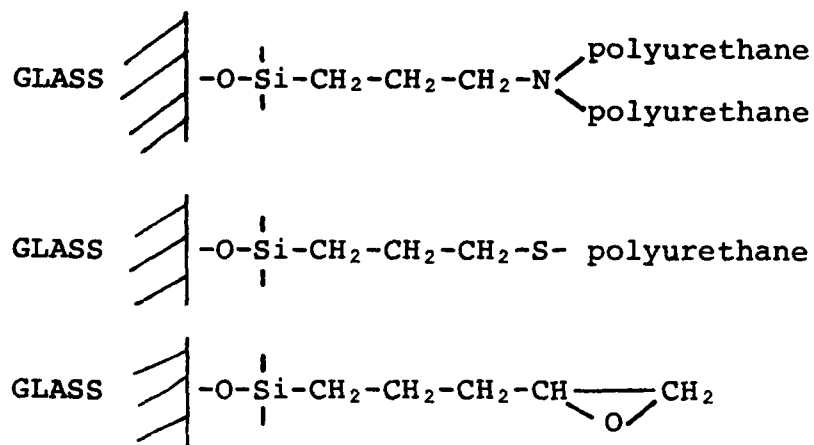


Fig. 2 - Effect of different treatments of glass surface on work of adhesion, W_A . ○ treated with AS, ● treated with MS, ○ treated with GS, ▲ abraded and treated with AS, □ abraded with #600 abrasive, + untreated.

with the -NCO group of the prepolymer, MS has just one reactive hydrogen atom and GS has none. The degree of enhancement of adhesion is related to the number of bonds that can form between the silane and polyurethane.



The glycidoxy group might react with the hydrogen atom that is present in the polyurethane backbone by a ring opening reaction but this reaction is much more difficult than the one between -NCO and -NH.

The wet adhesion was slightly improved for silane treated glass surfaces, but the durability of the joints still decreased considerably after soaking for several days. At equivalent concentrations of silane, the order of decrease in the work of adhesion was AS > GS > MS.

The polyurethane based on polybd resin R-45HT has good hydrolytic stability. Thus the losses in the work of adhesion after immersion in water must result from the adverse effect of

water on the interface rather than on the bulk properties of the polyurethane. Thermodynamic calculations indicate that if only secondary forces are acting across a glass/adhesive interface, water will virtually always desorb an organic adhesive from the glass surface¹⁰. In our experiments the presence of ionic bonds can result in markedly reduced adhesion in the presence of water, which would solvate the ionic clusters¹¹. Also according to Plueddeman¹², a flexible polymer like polybutadiene cannot form water resistant bonds to a hydrophilic mineral surface even with added silane coupling agents. As individual siloxane bonds are hydrolyzed, the silanols retract from the surface and are no longer available for new bond formation. Water gradually interposes itself throughout the interface until adhesion is completely lost.

From our observations, it seems that silane coupling agents do not exclude water from the interface but somehow function to retain adhesion in the presence of water. As shown in Fig. 2 at equivalent concentrations of silane, the order of decrease in the work of adhesion was AS > GS > MS. With the GS or MS treated glass surface, a fully condensed polysiloxane has some polarity arising from the terminal epoxy or mercapto groups, but these groups do not usually take part in hydrogen bond formation. The cured material would not be expected to show highly polar interaction with water. In the case of AS, the terminal amine group has a strong tendency to absorb water by hydrogen bonding and that could

lead to bond rupture and the poorer wet adhesion, compared to GS or MS pretreated glass surfaces, observed for AS pretreated glass surfaces.

When AS treated glass slide samples were prepeeled to some extent, a thin layer of elastomer was left on the glass surface due to cohesive failure. Upon immersion in distilled water for several days, the adhesion between the thin layer and the glass' surface was completely lost but the thin layer remained firmly attached to the elastomer. The initial W_A of AS treated glass surfaces to the polyurethane was 1234 KN/M^2 ; after soaking for 155 hr in distilled water at room temperature the W_A became 88 KN/M^2 . When the specimens were dried in vacuum at room temperature for 4 days, the W_A increased to 1000 KN/M^2 . Similar partial recovery in strength of adhesive joints after drying has been reported previously for bonds between polyethylene and glass or steel¹² and between epoxy adhesives and aluminum¹³. Permanent loss in bond strength has been ascribed to covalent bond rupture¹³ whereas recoverable loss in bond strength is variously attributed to reforming of hydrogen bonds¹³ and an equilibrium of bonding and debonding in the presence of water¹². In the present case strength recovery can be attributed to reformation of ionic bonds.

Adhesion of polyurethane to abraded glass surfaces

The durability studies of Gledhill and Kinloch support the conclusions reached above¹⁴. They conclude that in order to improve durability of adhesive joints to water either the water must be

prevented from reaching the interface or water resistant interfacial bonds must be formed. Formation of such bonds requires that special care should be taken to prepare the surface prior to joint preparation. Abrading a surface could be advantageous in two ways. First, the surface area would be increased and provide more sites for formation of chemical bonds. Second, a suitably abraded surface could provide sites for mechanical interlocking of the adhesive to the substrate. In the present study, grinding a glass surface with #600 abrasive from Barnes Engineering Company, gave high dry adhesion but had no effect on improving wet adhesion. (See Figure 2) If the ground glass surface was treated with AS before applying the polyurethane, the durability of the joint in the presence of water was much improved. Anchoring of the organic polymer in a rough surface improved dry adhesion but it was not effective at improving wet adhesion. When there were micropits along with an increased number of active chemical groups on the substrate surface as in the case of the AS pretreated ground glass surface, the wet adhesion was much improved.

Adhesion of chrome steel to polyurethane

Figures 3 and 4 show that after immersion in distilled water for several days the W_A of a petroleum ether cleaned chrome steel surface to polyurethane was decreased markedly. When the metal surface was pretreated with GS, CS, or MS, the treatments were similarly ineffective at improving the durability of the joints in water. (See Figure 3). Pretreatment

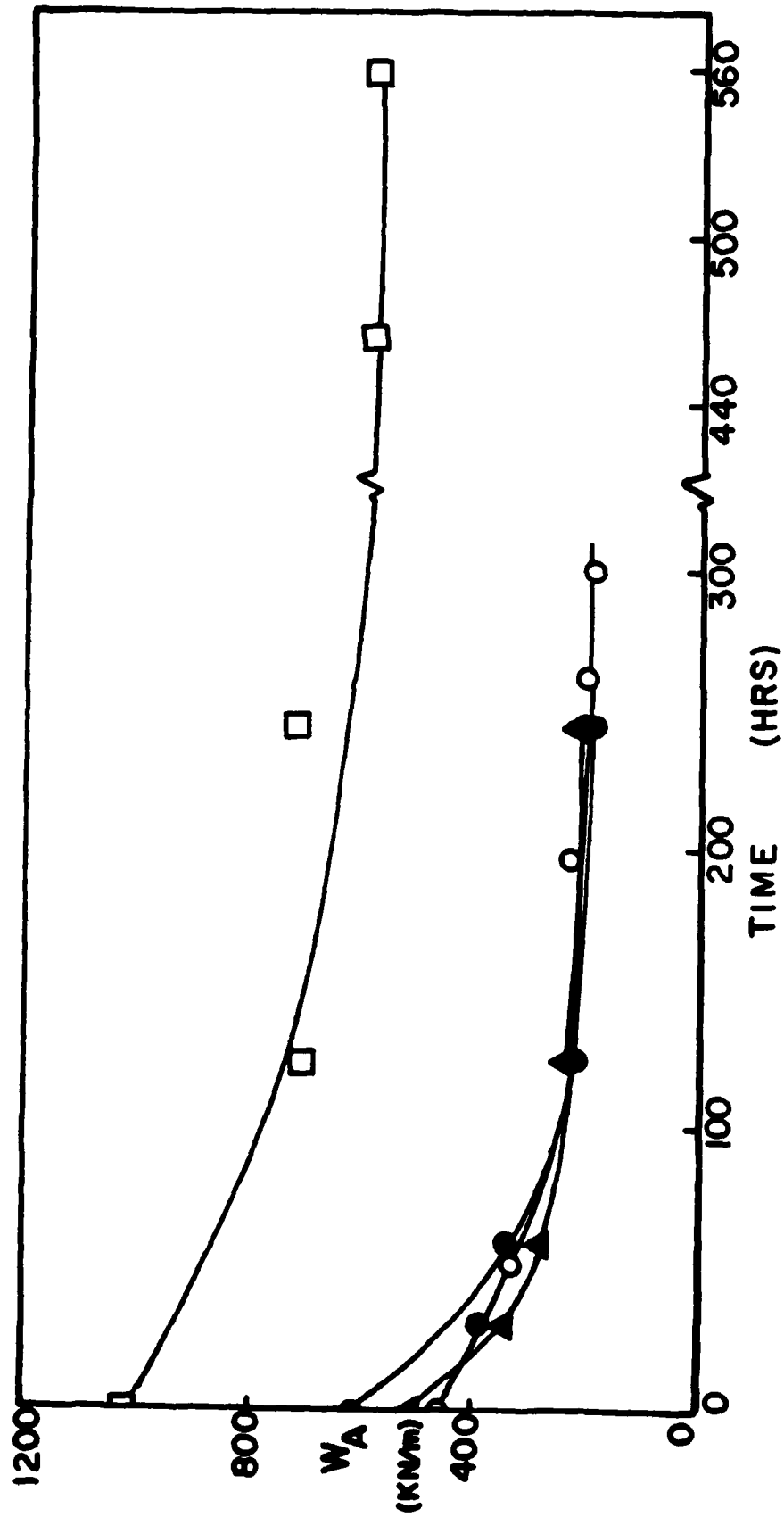


Fig. 3 - Effect of silane treatment of chrome steel surfaces on W_A .

\square treated with AS, \bullet treated with MS, \circ treated with GS,

\blacktriangle treated with CS.

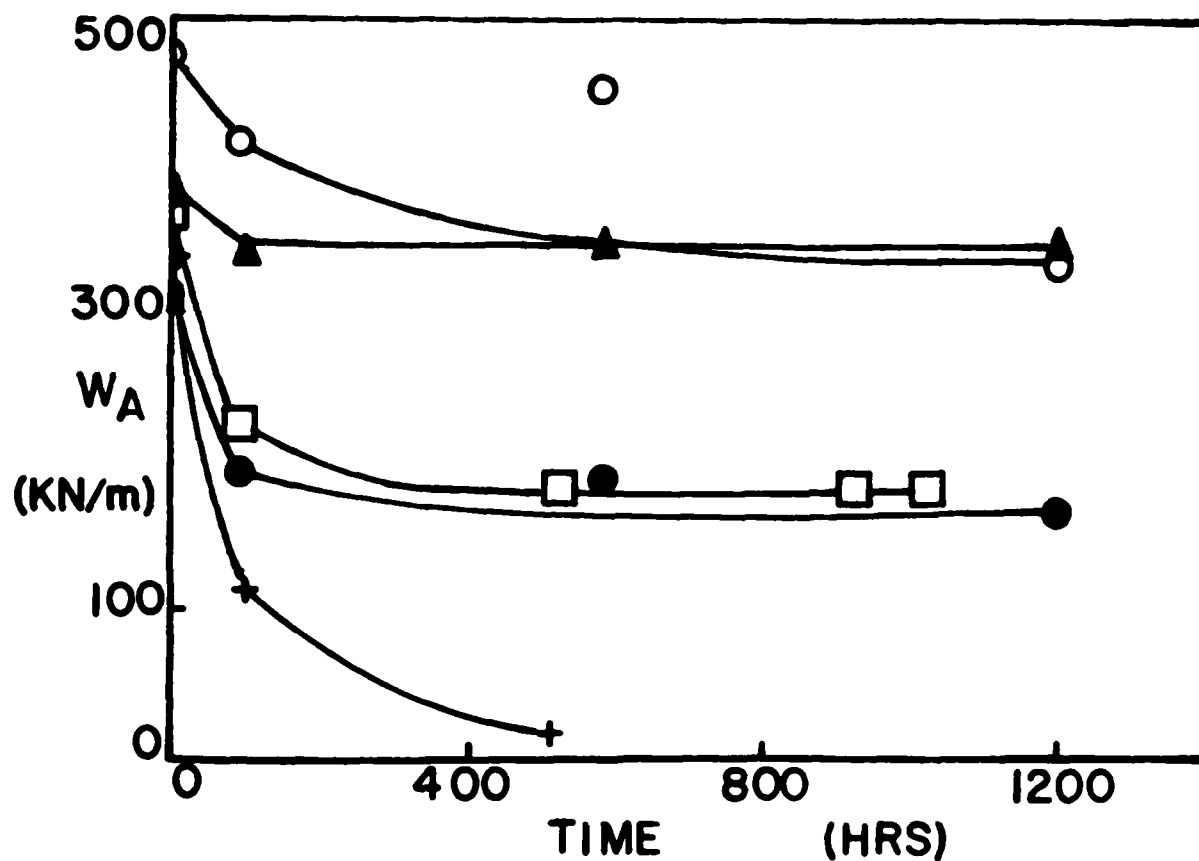


Fig. 4 - Effect of acid etching of chrome steel surface on W_A .

O etched with $\text{HCl} \cdot \text{HF} \cdot \text{H}_3\text{PO}_4$ (method 4), \blacktriangle etched with $\text{H}_2\text{C}_2\text{O}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ (method 3) and treated with prepolymer containing 0.5% GS, \square etched with $\text{H}_2\text{C}_2\text{O}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ (method 3), \bullet etched with $\text{HCl} \cdot \text{H}_2\text{O}$ (method 2), $+$ cleaned with petroleum ether (method 1).

with AS was very effective. The AS treated chrome steel surface gave a high W_A , even after immersion in distilled water for 23 days; its W_A after 23 days was still higher than the initial W_A of surfaces pretreated with the other silanes. GS, MS, CS have less effect on the steel surface than AS. The amino group of the latter silane can be chemisorbed by metal oxide surfaces¹⁵, and contribute to the high W_A observed.

The influence of chrome steel surface preparation

Surface preparation of substrates for joining adhesively plays a dominant and most important part in the reliability of the finished product. The prime purpose for surface preparation is to develop a bonding surface that will result in an optimum bond and provide the best service protection possible in the expected service environment.

Etching. Chrome steel surfaces were subjected to the surface treatments 2, 3, and 4 described in the experimental section. The effect of different surface treatment on the work of adhesion W_A of a chrome surface to polyurethane is shown in Fig. 4. HCl/HF/H₃PO₄ treated chrome steel surface gave the highest W_A , HCl/H₂O and H₂C₂O₄/H₂SO₄/H₂O treatment had similar effects on the W_A . Their wet adhesion was much improved compared to the unetched surface, probably as a result of mechanical interlocking of the

surface and the adhesive. In addition chemical treatment may have altered the surface to increase its free energy and make it more receptive to the adhesive. The usual result of chemical surface treatments of metals with acids in the presence of air is metal oxide formation¹⁶.

Experimental observations support the above conclusions. The morphology of the etched chrome steel surfaces is shown in Figure 5. As a result of acid etching, the surface profile was actually quite rough on a micro scale. $\text{HCl}/\text{HF}/\text{H}_3\text{PO}_4$ etching resulted in more micro etch pits than other treatments did. The $\text{H}_2\text{C}_2\text{O}_4/\text{H}_2\text{SO}_4/\text{H}_2\text{O}$ etched surface was smooth and micropit free. Probably the etching condition was too mild.

Chemical surface treatment not only produces micro etching pits but also increases the chemical polarity of the surface. Contact angles for water and glycerol on the etched metal surfaces are shown in Table 1. Chromium is resistant to the attack of a wide variety of chemicals at normal temperatures but it can react with many of these at elevated temperature^{10,17}.

When the $\text{H}_2\text{C}_2\text{O}_4/\text{H}_2\text{SO}_4/\text{H}_2\text{O}$ treated surface was pretreated with GS before applying the elastomer, the wet adhesion was

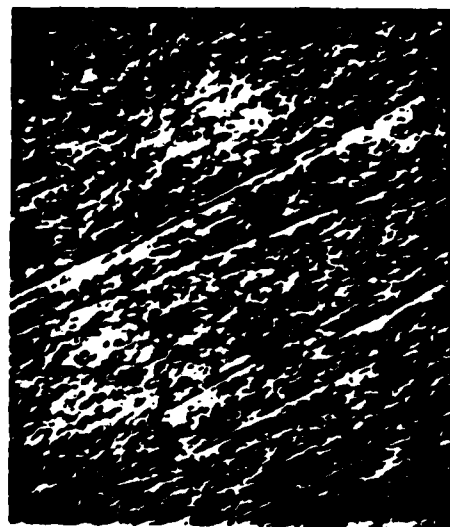
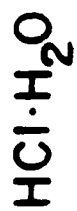
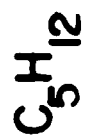


Fig. 5 - Electronmicrographs of etched chrome steel surfaces.

TABLE I

Contact Angle on Metal Surfaces

<u>Method of Surface Treatment</u>	<u>Contact Angle</u>	
	<u>Water</u>	<u>Glycerol</u>
<u>Chrome Steel</u>		
Cleaned with petroleum ether	51	52
Etched with $\text{HF} \cdot \text{HCl} \cdot \text{H}_3\text{PO}_4$	31	55
Etched with $\text{HCl} \cdot \text{H}_2\text{O}$	28	54
Etched with $\text{H}_2\text{C}_2\text{O}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$	34	46
<u>Aluminum</u>		
Cleaned with petroleum ether	60	44
Etched with 15% NaOH	49	35
Etched with $\text{Na}_2\text{CrO}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$	50	28
Etched with 15% NaOH and $\text{Na}_2\text{CrO}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$	52	29

enhanced significantly. The reasons for the improvement are probably similar to those given above in the section dealing with silane treatment of glass surfaces.

Etching of chrome steel plus silane in the resin recipe.

Silane coupling agents are usually applied to a substrate surface as a pretreatment from dilute aqueous solution. In this part of our study silane was added to the prepolymer at a concentration of 0.5g silane/100g prepolymer. The prepolymer containing silane and glycol was applied to a HF/H₃PO₄/HCl etched chrome steel surface. The effect of different silanes on the work of adhesion is shown in Fig. 6. The durability of the joint in the presence of water was much improved. Even after immersion in distilled water for over two months, there was only a slight decrease in W_A . The improved durability is attributed to increasing the extent of covalent bond formation between silane and acid etched metal surfaces compared to unetched surfaces. Polyurethane containing 0.5% GS gave the highest W_A and that containing 0.5% MS provided the lowest W_A . Probably part of the MS was consumed by reaction of -SH with NCO. The CH₂CH₂CH₂Cl group might to some extent form hydrogen bonds with the tertiary amine groups in the backbone of polyurethane, decrease the diffusion of GS into the interface, and result in the intermediate strength of adhesion.

The adhesion of aluminum to polyurethane

Aluminum is always covered with an oxide layer. The bonding interface of aluminum is not the metal itself but a variety of

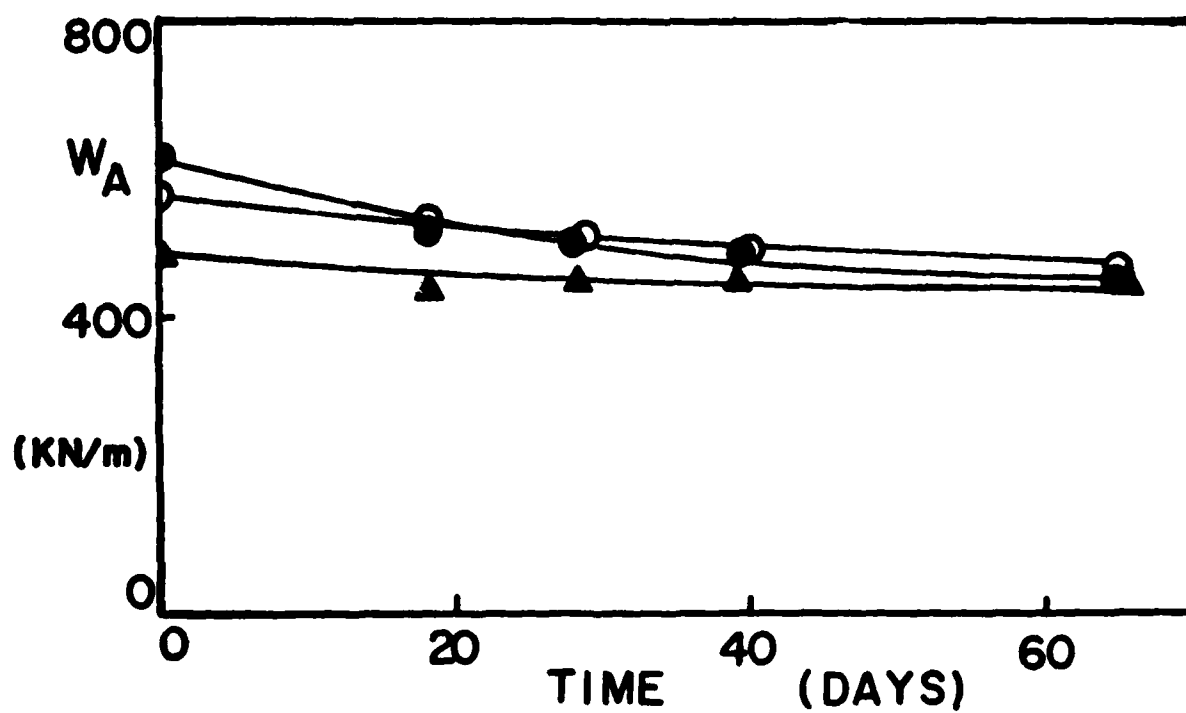


Fig. 6 - Effect of silane treatment of $\text{HF} \cdot \text{HCl} \cdot \text{H}_3\text{PO}_4$ etched chrome steel surfaces on W_A . ● treated with GS, ○ treated with CS, ▲ treated with MS.

aluminum oxide structures whose physical and chemical constitution can be altered through a variety of chemical procedures¹⁰. Bondability depends on the structure of the oxide layer. Water is absorbed on the surface of nonhydroscopic oxides like (Al_2O_3) as hydroxyl groups (M-OH) and as molecular water held by H-bonding to the surface hydroxyls.

Etching of aluminum. One of the methods to assess the quality of a pretreated surface is to make an adhesive joint. The work of adhesion, W_A , of different pretreated aluminum surfaces to polyurethane is shown in Fig. 7. Abraded aluminum surfaces gave high initial W_A , but during immersion in distilled water the rate of loss of W_A was similar to petroleum cleaned aluminum. Aluminum treated with aluminum surface treatment No. 8 gave much better wet adhesion than the others.

Figure 8 shows the change of morphology at different stages of pretreatment. The "as received" state of the aluminum surface is not structured but it is contaminated. After cleaning with petroleum ether the "clean" surface still has some loose particles (Fig. 8A). The alkaline degreased surface shows an irregular structure with pits and some loose particles on the surface (Fig. 8B). A $\text{Na}_2\text{CrO}_4/\text{H}_2\text{SO}_4$ etched surface gave a smooth etched surface with micro pits (Fig. 8C) and the alkaline cleaned-acid etched surface provided a concave, hilly micro morphology with oxide pits (Fig. 8D). In our experiments acid etching was influenced by alkaline pre-cleaning. A smooth surface gave a low peel strength whilst a

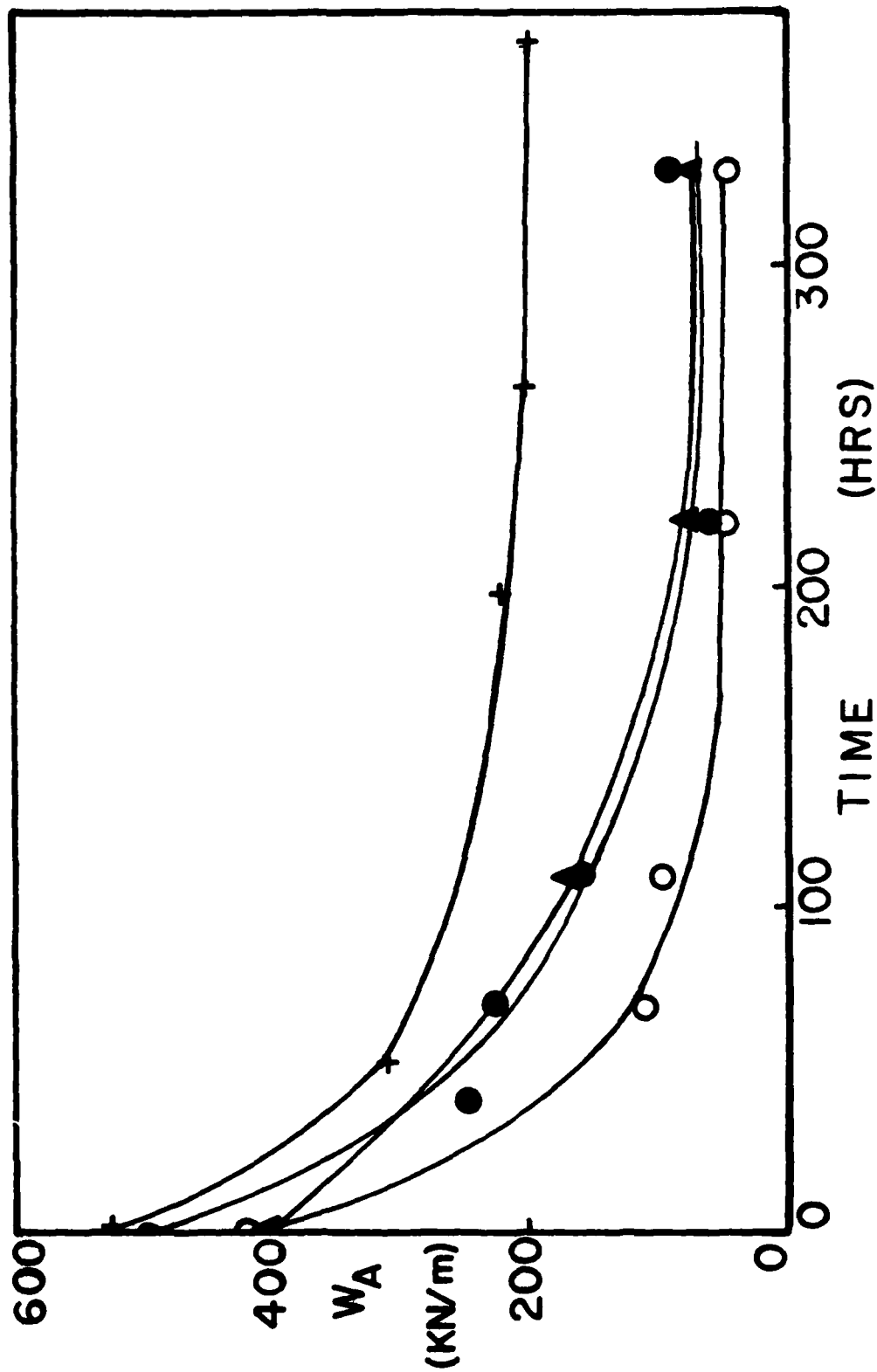
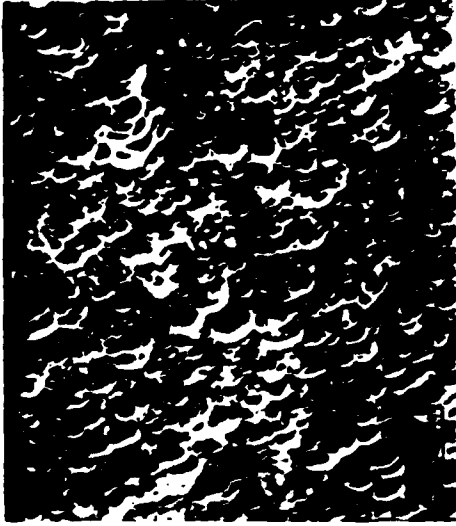


Fig. 7 - Effect of etching of aluminum on W_A + etched with 15% NaOH,

Δ etched with $\text{Na}_2\text{Cr}_2\text{O}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ (method 8), \bullet brushed with sand paper (method 6),

\circ etched with $\text{Na}_2\text{CrO}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ (method 7), \circ cleaned with petroleum ether (method 5).


 C_5H_{12}

 $15\% \text{NaOH}$

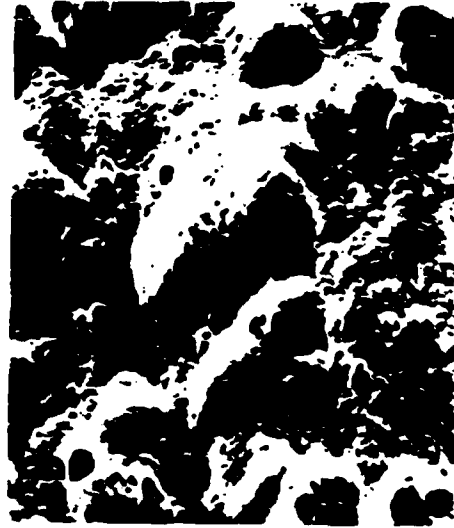
 $\text{Na}_2\text{CrO}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$

 $\text{NaOH}, \text{Na}_2\text{CrO}_4 \cdot \text{H}_2\text{SO}_4 \cdot \text{H}_2\text{O}$

Fig. 8 - Electronmicrographs of surface aluminum at different stages of pretreatment.

microscopically rough surface provided high peel strength. It seems the anchoring of the adhesive plays an important role.

Acid etching plus silane pretreatment. Bond failure in the presence of water is due to adsorption of water on the aluminum surface. In order to improve the durability of bond strength, in the presence of water, acid etched aluminum surfaces were treated with 0.5% silane. The effect of different silanes on the work of adhesion, W_A , of aluminum to polyurethane is shown in Fig. 9. Their wet adhesion was much improved; even after immersion in 70°C distilled water for 264 hrs, W_A decreased just slightly. At equivalent concentrations of silane, the order of durability of bond strength in the presence of water was AS > MS > GS. Trialkoxysilane chemically bonds to polyurethane by reaction of the organic functional group such as $-NH_2$, $-SH$ etc., with an $-NCO$ group of the prepolymer and may form metalosiloxane bonds to the aluminum surface. The degree of durability of the adhesion bond is related to the number of bonds that can form between the silane and polyurethane.

Base cleaning plus acid etching plus silane pretreatment of aluminum. The influence of alkaline cleaning in addition to acid etching is given in Fig. 10. Alkaline cleaned-acid etched aluminum surfaces gave higher W_A 's than just acid etched surfaces. Durability was also better. As before the reasons for the observed behavior probably include both improved mechanical interlocking and a greater number of interfacial covalent bonds.

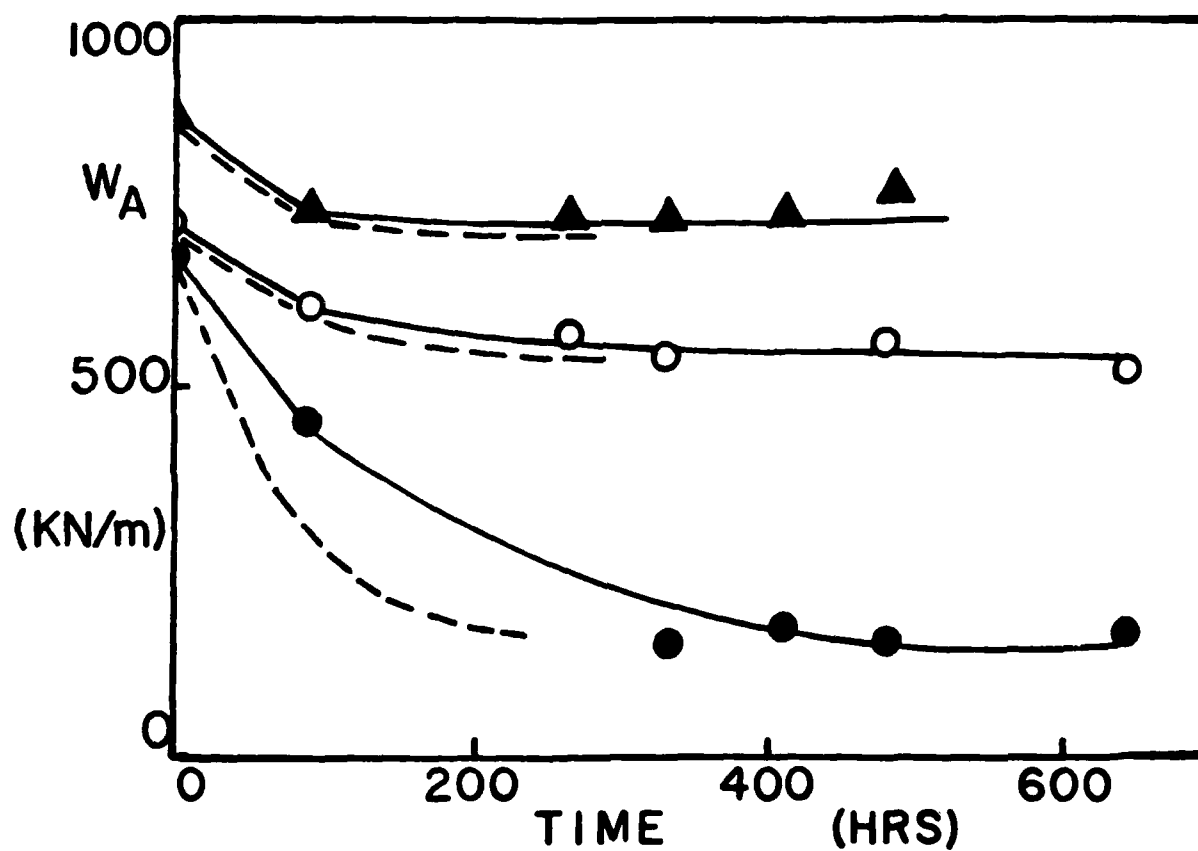


Fig. 9 - Effect on W_A of silane treatment on acid etched aluminum surfaces. \blacktriangle treated with AS, \circ treated with MS, \bullet treated with GS. — 25°C, --- 70°C.

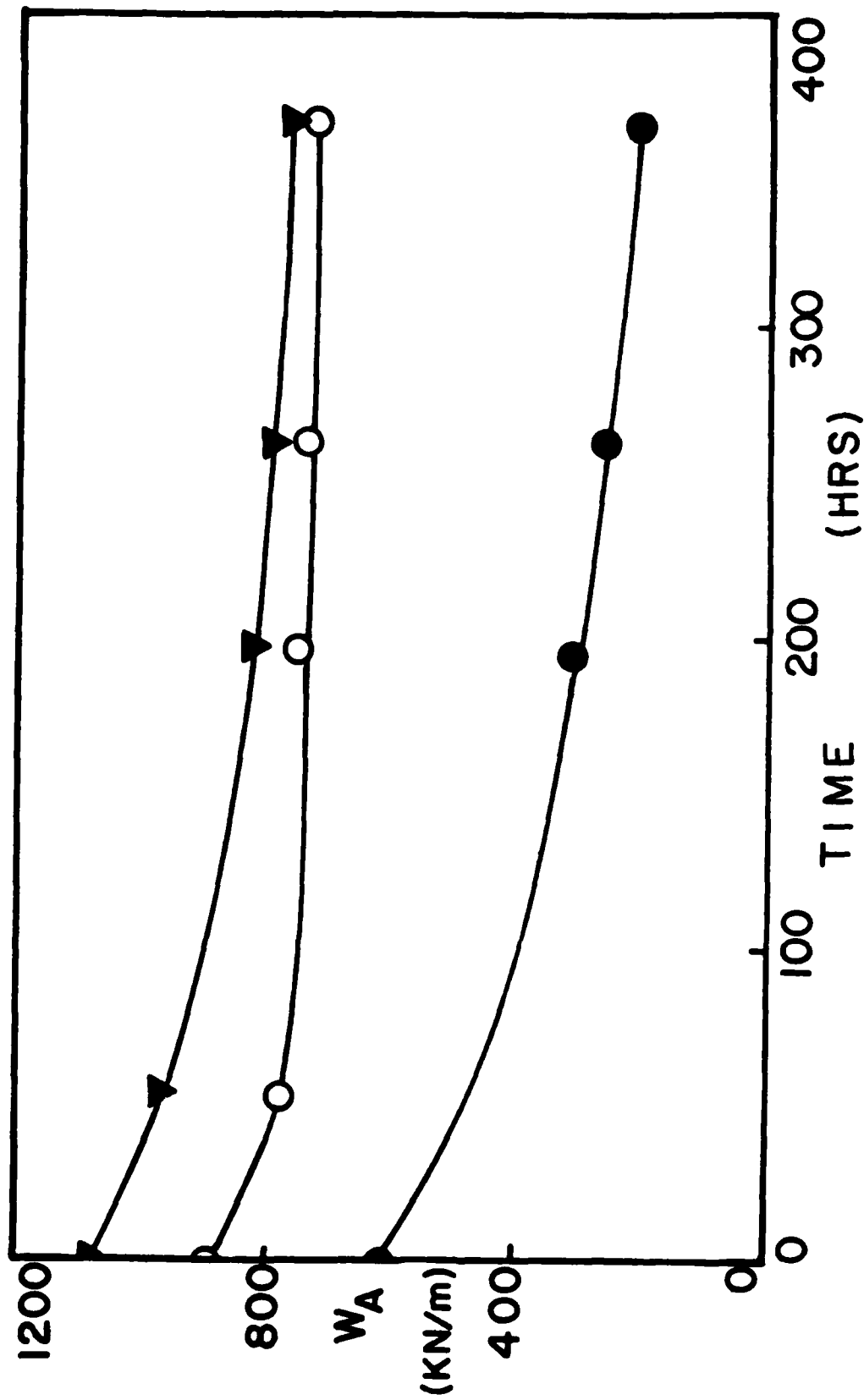


Fig. 10-Effect on W_A of silane treatment of NaOH cleaned and acid etched aluminum surfaces. ▲ treated with AS, ○ treated with MS, ● treated with GS.

Conclusion

The water durability of joints between glass or metal substrates and polyurethane can be greatly improved by proper surface preparation combined with the use of silane pretreatments. 3-Aminopropyltriethoxysilane is a very effective primer.

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References

1. F. Liang and P. Dreyfuss, Organic Coatings and Applied Polymer Science Proceedings, 48, 18 (1983).
2. "Polybd^R liquid Resin in urethane elastomers", product bulletin B.D. 3, ARCO Chemical Co., October, 1974.
3. "Polybd^R resin", General Bulletin, ARCO Chemical Co., January, 1978.
4. E. Warburg and T. Ihmori, Ann. d. Phys. 27, 481 (1886).
5. T. H. Elmer in "Silylated Surfaces", D. E. Leyden and W. Collens, Eds., Gordon and Breach Science Publishers, N.Y., 1980, p. 11.
6. Y. Eckstein and P. Dreyfuss, J. Adhesion, 13, 303 (1982).
7. Y. Eckstein and P. Dreyfuss, J. Adhesion, 15, 163 (1983).
8. Y. Eckstein and P. Dreyfuss, J. Adhesion, 15, 179 (1983).

9. Y. Eckstein and P. Dreyfuss, J. Adhesion, 15, 193 (1983).
10. A. J. Kinloch, W. A. Dukes and R. A. Gledhill, in "Adhesion Science and Technology", L. H. Lee, Ed., Plenum Press, N.Y., N.Y. 1975, p. 597.
11. R. Rembaum, in "Advances in Urethane Science and Technology", 2, K. K. Frisch and S. L. Reegen, Eds, Technomic Pub. Co. Stanford, Conn., 1971, p. 109.
12. E. P. Plueddemann, J. Adhesion, 2, 184 (1970).
13. R. I. Butt and J. L. Cotter, J. Adhesion, 8, 11 (1976).
14. R. A. Gledhill and A. J. Kinloch, Adhesion, 1974, 315.
15. T. Furakawa, N. K. Eib, K. L. Mittal, H. R. Anderson, Jr., SIA Surf. Interface Anal., 4(6), 240 (1982).
16. C. V. Cagle, "Adhesive Bonding", McGraw Hill Book Co., N.Y., N.Y. 1968, p. 87.
17. J. C. Bailar, H. J. Emeléus, Sir Ronald Nyholm and A. F. Trotman-Dickenson, Eds., "Comprehensive Inorganic Chemistry", 3, Pergamon Press, distributed by Compendium Publishers, Elmsford, N.Y., 1973.
18. J. D. Minford, Adhesives Age, March, 1978, p. 17.

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